Introduction

The movement towards greener alternatives in polymer science steadily grows as the drive for sustainability continues. The increasing concerns regarding environmental pollution and crude oil depletion and price fluctuation have pulled both academia and industry to focus more on green raw materials, chemistry, and processing. To this regard, the conversion of renewable resources into prevalent polymer materials through enzymatic polymerization has become a particularly irresistible path. Compared to conventional chemically-catalyzed processes, enzymatic polymerization has become proven to be effective as a more eco-friendly synthesis route. In addition to the mild reaction conditions, the high selectivity of enzymes also allows to avoid tedious protection–deprotection steps and improves the quality of the end products. In the past few years, enzymes have been used to synthesize a wide array of polyester classes, for instance, polyesters, polyamides, vinyl polymers, and polysaccharides. Nevertheless, compared to the conventional synthetic route, the application of enzymatic polymerization is somehow still economically limited. One interesting approach to circumvent this limitation is to design sustainable high-performance polymers for technologically relevant applications.

In general, aromatic compounds provide rigidity to a polymer chain, owing to the inhibition of rotational flexibility. Polymers with rigid backbones are often characterized by their high thermal and mechanical stability and are therefore suitable for the use as high-performance polymers. Among them, furan-based polyesters are promising sustainable alternatives with great interest. In addition to their sustainability, they possess similar or even better properties than their petrol-based counterparts. For example, poly(ethylene furanoate) (PEF) shows better barrier properties than poly(ethylene terephthalate) (PET).

An array of different furan polyesters has been synthesized by the groups of Okada, Ballauf, and Gandini since the 1990s. Thereafter, various furanic–aliphatic polyesters have been reported, such as poly(ethylene furanoate) (PEF), poly(butylene furanoate) (PBF), and poly(2,3-butylene furanoate) (P23BF). We recently found that enzymatic polymerization can be used to synthesize different semi-aromatic furan-based polyesters (Scheme 1), by using dimethyl 2,5-furandicarboxylate (DMFDCA) or 2,5-bis(hydroxymethyl) furan (BHMF).

To further enhance the properties, the additional incorporation of aromatic content through copolymerization, in which two or more different polyester backbones are chemically linked together, can be an interesting approach. For example, Ma et al. applied various conventional methods to synthesize furan-based copolyesters. Recently, Morales-Huerta et al. reported the enzymatic ring opening polymerization of poly(butylene 2,5-furandicarboxylate-co-butylene succinate) and poly(r-caprolactone-co-butylene 2,5-furandicarboxylate).

Inspired by our previous findings, we explored the enzymatic copolymerization of two carbohydrate-sourced monomers (DMFDCA and BHMF) with aliphatic linear monomers, to prepare several semi-aromatic copolyesters. By performing a detailed analysis of the enzymatic copolymerization, we observed...
the distinct activity of the enzyme towards different building blocks. We also investigated their morphologies, as well as the thermal properties of the obtained furan-based copolyesters.

Results and Discussion

Synthesis and structural characterization

Furan-based copolyesters were synthesized by a two-step temperature-varied enzymatic polymerization. The enzymatic copolymerization followed two different synthesis approaches, as depicted in Scheme 2. In the first approach, the furan-based copolyesters were prepared by using DMFDCA, BHMF, and an aliphatic linear diol as the building blocks, whereas in the second approach, linear diacid ethyl esters were used. The number of the methylene units \(n\) in the dicarboxylic segments of the diacid ethyl esters is 2, 4, 6, 8, or 10, whereas the in aliphatic linear diols, \(n\) is 4, 6, 8, 10 or 12. In this study, this number is defined as the chain length of the tested aliphatic linear monomers. The obtained furan-based copolyesters are listed in Table 1.

The chemical structures and compositions of the copolyesters were confirmed by ATR-FTIR and \(^1\)H-NMR spectroscopy (Figure 1). The ATR-FTIR and \(^1\)H NMR spectra of the representative furan-based copolyesters from DMFDCA, BHMF, and diacid ethyl esters are illustrated in Figure S1 (see the Supporting Information). Detailed NMR and IR peak assignments are available in the Experimental Section. The molecular weights, yield, and the monomer feed compositions are summarized in Tables S1 and S2.

Influence of aliphatic linear monomers on the enzymatic copolymerization of the furan-based copolyesters

To evaluate the influence of aliphatic linear monomers on the enzymatic synthesis of the furan-based copolyesters, a comparative study on the degree of polymerization of the whole series of the furan-based copolyesters was performed.

To study the effect of the chain length of aliphatic linear diols, all furan-based copolyesters obtained from the first approach were evaluated (Figure 2 and Table S1). The results indicate that *Candida antarctica* lipase B (CALB) prefers longer linear diols \((n=8,10\) and 12) compared to shorter linear diols \((n=4\) and 6). If 1,8-ODO was used, the enzymatic polymerization resulted in P(FMF-co-OF) with a number-average degree of polymerization \((\text{DP}_n)\) of 122 and a weight-average degree of polymerization \((\text{DP}_w)\) of 269, which was the highest amongst the tested aliphatic diols. Furan-based copolyesters with relatively similar \(\text{DP}_n\) and \(\text{DP}_w\) values could be obtained by using 1,10-ODO and 1,12-DODO. Upon decreasing the chain length to 6 and 4 (1,6-HDO and 1,4-BDO), the \(\text{DP}_n\) and \(\text{DP}_w\) values of furan-based copolyesters were decreased significantly. These results corroborate our previous finding on the preference of CALB on longer chain lengths of aliphatic linear diols \[^{[5d]}\].

From the second synthetic approach, the same \(\text{DP}_n\) and \(\text{DP}_w\) trend is observed with respect to the diacid ethyl ester chain length. Furan-based copolyesters with the highest \(\text{DP}_n\) and \(\text{DP}_w\) values of 73 and 137 were obtained by using diethyl adipate \((n=4\); Figure 2 and Table S2). By increasing the diacid ethyl ester length to \(n=6\) and 8, furan-based copolyesters

![Scheme 1](image1.png)

Scheme 1. Enzymatic synthesis of semi-aromatic furan-based polyesters from (a) DMFDCA and aliphatic diols and (b) BHMF and diacid ethyl esters.

![Scheme 2](image2.png)

Scheme 2. Enzymatic synthesis of furan-based copolyesters/co-oligoesters from (a) DMFDCA, BHMF, and aliphatic diols and (b) DMFDCA, BHMF, and diacid ethyl esters by a two-stage method in diphenyl ether.
with relatively similar $\text{DP}_n$ and $\text{DP}_w$ values were obtained. However, if diethyl succinate ($n=2$) and diethyl dodecanedioate ($n=10$) were used as the monomer, the resultant furan-based copolyesters had only very low $\text{DP}_n$ and $\text{DP}_w$ values. Similar results on the effect of diacid ethyl ester/dicarboxylic acid chain length on enzymatic polymerization were reported previously.

This result can be explained by the variable specificity of CALB towards diacid acyl esters with different chain length. This explanation is also in agreement with the study reported by McCabe and Taylor on the acyl-binding site of CALB. They found that adipic acid is the preferred substrate among the tested dicarboxylic acids, which is owing to its low entropic component contribution to the enantioselectivity of CALB.

Interestingly, by changing the aliphatic monomers from aliphatic diols to diacid ethyl esters, enzymatic polymerization, in general, resulted in copolyesters with significantly lower DP. This can be explained by the instability of BHMF, which results in either formation during the polymerization. As we reported previously, the high reactivity of the OH group in BHMF can lead to dehydration or reaction with ethanol to form BHMF ethers. Consequently, the copolyester chain propagation will be greatly limited by the formation of BHMF ethers as chain stops. The formation of BHMF ether was further confirmed by the presence of a small peak ($\leq 1\text{ wt\%}$) at $\delta \approx 4.40$ ppm in the $^1\text{H}$ NMR spectra, as we reported previously. However, the substrate specificity of the enzyme also should be taken into account and this will be discussed further in later sections.
Effect of monomer feed composition on enzymatic synthesis of the furan-based copolyesters

One fundamental issue in this work was to understand the monomer incorporation mechanism during the copolyester formation, which seemed to be governed by the enzyme catalytic activity and consequently influenced the molecular weights of the resulting furan-based copolyesters. In this study, various molar feed compositions were used and evaluated.

We observed that as soon as we increased the 2,5-furandi-methylene furanocate (FMF) molar feed fraction to 50%, the DPw and DPn of P(FMF-co-DOF) significantly decreased from 86 to 11 and 14, respectively. A similar trend was also observed in other copolyester series (Table 2). These results imply that the propagation mechanism is not solely limited by the formation of BHMF ethers in the system. Substrate specificity of the enzyme can also determine the enzyme catalytic activity and may also be an explanation for this. Like many enzymes, depending on the structural complementarity of transition state with the active site, CALB has the capability to catalyze diverse reactions at different efficiency ranges.[23]

A possible copolymerization mechanism of P(FMF-co-DOF) is depicted in Scheme 3. In this mechanism, the polymerization starts with the formation of the acyl–enzyme complex and continues with polycondensation. We propose that during the polycondensation, an intermediate product (b) forms that can inhibit the polymerization. Steric hindrance of (b) creates structure incompatibility with the enzyme active site; consequently, the polymer growth is terminated. Another possible explanation is that the OH functionality in (b) transforms into ethers, as in the case of the BHMF and eventually terminates the copolyester chain elongation. The proposed copolymerization mechanism appears to be well substantiated by the constantly lower value of the FMF molar fraction in the copolyester segment compared to the corresponding feed. Additionally, Takwa previously reported similar findings regarding the low activity of CALB towards d,l-lactide,[23] owing to the bulky conformation of the lactide if acylated. However, future studies by molecular modeling are recommended to validate the proposed reaction mechanism.

### Table 2. Molar fraction and degree of polymerization of the furan-based copolyesters obtained from different feed compositions of DMFDCA, BHMF, aliphatic diols, and diacid ethyl esters.

<table>
<thead>
<tr>
<th>Feed (%)</th>
<th>Molar Fraction [%]</th>
<th>Copolyester</th>
<th>DPw</th>
<th>DPn</th>
</tr>
</thead>
<tbody>
<tr>
<td><strong>FIRST APPROACH</strong></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-BF)</td>
<td>25</td>
<td>75</td>
<td>16</td>
<td>84</td>
</tr>
<tr>
<td>P(FMF-co-HF)</td>
<td>25</td>
<td>75</td>
<td>23</td>
<td>77</td>
</tr>
<tr>
<td>P(FMF-co-OF)</td>
<td>25</td>
<td>75</td>
<td>22</td>
<td>78</td>
</tr>
<tr>
<td>P(FMF-co-DF)</td>
<td>25</td>
<td>75</td>
<td>22</td>
<td>78</td>
</tr>
<tr>
<td>P(FMF-co-DOF)</td>
<td>25</td>
<td>75</td>
<td>22</td>
<td>78</td>
</tr>
<tr>
<td><strong>SECOND APPROACH</strong></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-FMS)</td>
<td>25</td>
<td>75</td>
<td>50</td>
<td>70</td>
</tr>
<tr>
<td>P(FMF-co-FMA)</td>
<td>25</td>
<td>75</td>
<td>50</td>
<td>70</td>
</tr>
<tr>
<td>P(FMF-co-FMSu)</td>
<td>25</td>
<td>75</td>
<td>50</td>
<td>70</td>
</tr>
<tr>
<td>P(FMF-co-FMSe)</td>
<td>25</td>
<td>75</td>
<td>50</td>
<td>70</td>
</tr>
<tr>
<td>P(FMF-co-FMMD)</td>
<td>25</td>
<td>75</td>
<td>50</td>
<td>70</td>
</tr>
</tbody>
</table>

[a] $F_{FMF}$ and $F_{XF}$ represent the molar feed ratios of PFMY, PXF (in the first approach), and PFMX (in the second approach), respectively. [b] $X_{FMF}$ and $X_{XF}$ represent the molar fractions of PFMY, PXF (in the first approach), and PFMX (in the second approach) segments in the obtained furan-based copolyesters, respectively, determined by $^1$H NMR spectroscopy. [c] $X_{FMF}$, [d] $X_{XF}$, and $X_{FMX}$ represent the molar fractions of PFMY, PXF (in the first approach), and PFMX (in the second approach) segments in the obtained furan-based copolyesters, respectively, determined by $^1$H NMR spectroscopy. [e] $X_{FMF}$, [f] $X_{XF}$, and [g] $\bar{X}_{P}$ (number-average degree of polymerization) = $2 \times (\frac{\overline{M}_{P} - 62.06}{\overline{M}_{P}}) / \{X_{FMF} \times \overline{M}_{FMF} \times \overline{M}_{P} \} + \{X_{XF} \times \overline{M}_{XF} \times \overline{M}_{P} \} + \{X_{FMX} \times \overline{M}_{FMX} \times \overline{M}_{P} \}$; [h] $\bar{X}_{w}$ (weight-average degree of polymerization) = $2 \times (\frac{\overline{M}_{w} - 62.06}{\overline{M}_{w}}) / \{X_{FMF} \times \overline{M}_{FMF} \times \overline{M}_{w} \} + \{X_{XF} \times \overline{M}_{XF} \times \overline{M}_{w} \} + \{X_{FMX} \times \overline{M}_{FMX} \times \overline{M}_{w} \}$; [i] Not determined.

Thermal properties of furan-based copolyesters

The thermal stability of the obtained furan-based copolyesters was characterized by TGA and the representative characteristic curves for their thermal degradation behaviors are depicted in Figure 3. The values of the degradation temperatures are summarized in Table 3. The copolyesters from the first synthetic approach showed TGA traces with a two-step degradation pattern (Figure 3a). It consists of an initial degradation at around 230–250 °C with only approximately 10% weight loss, followed by a secondary degradation with a maximum degradation temperature at approximately 390 °C.

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In contrast, the thermal degradation profiles of the furan-based copolyesters from the second approach show the polymers to be less stable (Figure 3b). Their degradation occurs in several stages and a significant weight loss was detected at temperatures around 220–280 °C. To conclude, the furan-based copolyesters obtained from the first synthetic approach appeared to have better thermal stability, which is mainly owing to their higher molecular weights. It should be noted that, for copolyesters obtained from the first synthetic approach, the chain length of the tested aliphatic diols and molar compositions of the monomeric units have no significant influence on the decomposition temperature of the resulting copolyesters. Furthermore, the tested furan-based copolyesters have similar degradation profiles and temperatures to their furan-based polyester counterparts, as we reported previously. 

Table 3. Thermal properties of the obtained furan-based copolyesters.

<table>
<thead>
<tr>
<th>Copolyester</th>
<th>DSC$^{[a]}$</th>
<th>Tg$^{[c]}$</th>
<th>Tm$^{[c]}$</th>
<th>Tc$^{[c]}$</th>
<th>Tcc$^{[c]}$</th>
<th>Td-$10%$</th>
<th>Td-$max$</th>
</tr>
</thead>
<tbody>
<tr>
<td>P(FMF-co-BF)$^{[a]}$</td>
<td>15</td>
<td>142</td>
<td>83</td>
<td>30</td>
<td>230</td>
<td>370</td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-HF)$^{[a]}$</td>
<td>12</td>
<td>120</td>
<td>71</td>
<td>240</td>
<td>340</td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-OF)$^{[a]}$</td>
<td>2</td>
<td>123</td>
<td>47</td>
<td>240</td>
<td>390</td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-DF)$^{[a]}$</td>
<td>6</td>
<td>90</td>
<td>30</td>
<td>250</td>
<td>390</td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-DOF)$^{[a]}$</td>
<td>2</td>
<td>88</td>
<td>61</td>
<td>240</td>
<td>390</td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-FMS)$^{[a]}$</td>
<td>-6</td>
<td>78</td>
<td>30</td>
<td>220/310/430</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-FMA)$^{[a]}$</td>
<td>-8</td>
<td>78</td>
<td>30</td>
<td>240</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-FMSe)$^{[a]}$</td>
<td>-16</td>
<td>57</td>
<td>30</td>
<td>250/310/460</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-FMD)$^{[a]}$</td>
<td>-19</td>
<td>62</td>
<td>30</td>
<td>250/310/450</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-FMSu)$^{[a]}$</td>
<td>-19</td>
<td>79</td>
<td>43</td>
<td>280/320/460</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>P(FMF-co-FMDO)$^{[a]}$</td>
<td>-19</td>
<td>79</td>
<td>43</td>
<td>280/320/460</td>
<td></td>
<td></td>
<td></td>
</tr>
</tbody>
</table>

[a] Furan-based copolyesters from DMFDCA, BHMF, and aliphatic diol with a feed ratio of 50:12.5:37.5. [b] Furan-based copolyesters from DMFDCA, BHMF, and diacid ethyl esters with a feed ratio of 12.5:50:37.5. [c] Tg = glass transition temperature from the modulated DSC heating scan; Tm = melting temperature from the second DSC heating scan; Tc = crystallization temperature upon cooling; Tcc = cold crystallization temperature from the second DSC heating scan. [d] Td-$10\%$ = decomposition temperature at 10% weight loss; Td-$max$ = temperature at the maximum rate of decomposition. [e] Not detected in the tested temperature range. [f] Measured from the first DSC heating scan.
To evaluate the thermophysical behavior, a comparative differential scanning calorimetry (DSC) study was performed. The thermal transitions of the whole copolyester series are listed in Table 3 and representative DSC and temperature-modulated DSC (TMDSC) traces of P(FMF-co-OF) is plotted in Figure 4. To enhance the visibility of the $T_m$ we performed the TMDSC measurement.

For the furan-based copolyesters obtained from the first synthetic approach, broad and multiple melting peaks are observed in all first heating cycles. A melting peak disappeared in the second heating scan of P(FMF-co-DF) and no crystallization was observed in the cooling cycle of P(FMF-co-HF) and P(FMF-co-DOF). In addition, a cold crystallization transition ($T_c$) can be observed before the melting temperature ($T_m$) in the second heating scan of P(FMF-co-OF) and P(FMF-co-DOF). This can be explained by the polymer molecular motion and orientation, which could act as nuclei and promote the spontaneous cold crystallization. In general, we observed a constant decrease in their $T_g$ and $T_m$ values as we increased the chain length of the aliphatic linear diol. The decrease in $T_g$ value is caused by the increasing chain flexibility and mobility provided by the longer aliphatic chains. The decrease in $T_m$ value is corroborated well with our previous results of the corresponding furan-based polyesters.

Except for P(FMF-co-FMD), no crystallization is detected in DSC traces of the furan-based copolyester series obtained from the second synthetic approach. Owing to their low crystallization rate, no crystallization is observed, and the $T_m$ disappeared in the second heating scan of P(FMF-co-FMSu) and P(FMF-co-FMSe). For P(FMF-co-FMS) and P(FMF-co-FMA), they are amorphous materials because no melting or crystallization was observed. To verify this, we conducted further analysis by wide-angle X-Ray diffraction (WAXD) measurements, which is discussed below.

Crystallinity of furan-based copolyesters
The WAXD spectra and POM images confirm the semicrystalline properties of the furan-based copolyesters obtained from the first approach (Figure 5). As shown in Figure 5a, P(FMF-co-HF), P(FMF-co-OF), P(FMF-co-DF), and P(FMF-co-DOF) gave rise to similar WAXD patterns, which displayed an amorphous halo at $2\theta \approx 22^\circ$ and three reflection peaks at $2\theta \approx 10.0–13.6^\circ$, $16.8–17.9^\circ$, and $24.1–24.9^\circ$. P(FMF-co-BF) has three similar diffraction peaks with one additional reflection peak at $22.5^\circ$.

Interestingly, we observed similar WAXD patterns in their furan-based polyester counterparts (PBF, PHF, POF, and PDF). This suggested that the furan-based copolyesters have similar crystal structures to their furan-based polyester counterparts and that they do not change significantly as we incorporate more aromatic content in the main chain. As represented in Figure 4b, the POM image of P(FMF-co-DOF) clearly showed that the product consists of birefringent spherulites with an estimated particle size of approximately 50 μm.

For the furan-based copolyesters obtained from the second approach, distinct morphologies were observed. Analogous to their polyester counterparts, P(FMF-co-FMSu), P(FMF-co-FMSe), and P(FMF-co-FMD) are all semicrystalline materials. They showed a similar WAXD pattern with six reflection peaks at $2\theta \approx 13.7–13.8^\circ$, $17.1–17.2^\circ$, $20.2–20.4^\circ$, $21.4–21.7^\circ$, $22.8–22.9^\circ$, and $23.7–23.8^\circ$ (Figure S3). In contrast, WAXD spectra of P(FMF-co-FMS) and P(FMF-co-FMA) only displayed a broad halo at $2\theta \approx 22^\circ$. This result agrees well with the DSC results, indicating that both P(FMF-co-FMS) and P(FMF-co-FMA) are indeed amorphous materials. These two copolymers, in particular, showed different morphologies with their polyester counterparts, namely PFMS and PFMA. As we incorporated FMF into the polymer main chain by copolymerization, the polymer morphology changed from semicrystalline to a completely amorphous structure.

Conclusions
We have shown that the application of enzymatic polymerization techniques can be extended to prepare a series of sustainable furan-based copolyesters with increased content of ar-
omeric units. Hereby, we introduce two different synthetic approaches. Using the first approach, a mixture of the furan monomers with an aliphatic diol yields a series of copolymers with comparable high molecular weights up to $M_n \geq 35\,000 \text{g mol}^{-1}$, which is advantageous for future processability. By changing the aliphatic monomers from diols to diacid ethyl esters (second approach), a significant decrease in the molecular weight was observed. This can be explained by BHM ether formation in the reaction system and the monomer incorporation mechanism during the copolymerization. This is further supported experimentally by the constant lower value of BHMF molar fraction in the copolymers compared to the corresponding feed ratio, regardless of the variation in BHMF feed ratio.

The thermal stability of all copolymers reported herein was established by TGA analysis, which indicates industrially relevant applicability. Compared to their furan-based polyester counterparts, they possess similar decomposition profiles. DSC analysis provides insights into the thermal behavior, especially with regard to the crystallization ability of the furan-based copolymers. Quite interestingly, aside from P(FMF-co-FMS) and P(FMF-co-FMA), all copolymers are semicrystalline materials. This suggests that the presence of the FMF segment in the main chain hinders the crystallinity of the furan-based copolymers.

One limitation of our research is that the molecular weight of the copolymers is restricted by the incorporation of aromaticity in the backbone. Future work should focus on enhancing the molecular weight, as well as increasing the aromatic content in the polyester chain. As such, we believe that this study provides a fundamental background to design sustainable high-performance polymers by an enzymatic pathway.

### Experimental Section

**Materials**

Novozym 435 (N435, Candida antarctica lipase B (CALB)) immobilized on acrylic resin, $\geq 5000 \text{U g}^{-1}$, 1,4-butanediol (1,4-BDO, 99 %), 1,6-hexanediol (1,6-HDO, 99 %), 1,8-octanediol (1,8-DDO, 98 %), 1,10-decanediol (1,10-DDO, 98 %), 1,12-dodecanediol (1,12-DDO, 99 %), diethyl succinate (99 %), diethyl adipate (99 %), diethyl sebacate ($\geq 98 %$), deuterated chloroform (CDCl$_3$, 99.8 atom % D), chloroform (CHCl$_3$, Chromasolv HPLC, $\geq 99.8 \%$, amylene stabilized) and diphenyl ether (99 %) were purchased from Sigma–Aldrich. Dimethyl 2,5-furan dicarboxylate (DMFDCA, 97 %) was purchased from Fluorochem UK. Diethyl dodecanedioate ($\geq 95\%$) was purchased from TCI Europe. Diethyl suberate (99 %) was purchased from ABCR. 2,5-Bis(hydroxymethyl)furan (BHMF, $\geq 98 \%$) was purchased from Apollo Scientific. Chloroform (CHCl$_3$, ChromAR HPLC, ethanol stabilized) and n-hexane (n-Hx, 99 %) were obtained from Macron. Absolute methanol (MeOH, AR) was obtained from Bio-Solve Chemicals. N435 was predried as reported previously.$^{[5b]}$ Diphenyl ether was distilled at 140 °C under reduced pressure and stored with activated 4 Å molecular sieves before use. All other chemicals were used as received.

**Instrumental methods**

Nuclear Magnetic Resonance (NMR) measurements were performed on a Varian VXR spectrometer (1H: 400; 13C: 300 MHz), using CDCl$_3$ as the solvent.

Attenuated total reflectance–Fourier transform infrared (ATR-FTIR) spectra were recorded on a Bruker VERTEX 70 spectrometer equipped with a Platinum-ATR diamond single reflection unit. The measurement resolution was 4 cm$^{-1}$ and the spectra were collected in the range of 4000–400 cm$^{-1}$, with 16 scans for each sample. Atmospheric compensation and baseline correction were applied to the collected spectra using OPUS spectroscopy software (v7.0) (Bruker Optics).

Size-exclusion chromatography (SEC) was performed on a Malvern Viscotek GPCmax equipped with triple detection, consisting of a Malvern Dual detector and Schambek RI2912, refractive index detector. The separation was performed by utilizing two PLgel 5 µm MIXED-C, 300 µm columns from Agilent Technologies at 35 °C. Amylene-stabilized chloroform (CHROMASOLV, for HPLC, $\geq 99.8 \%$) was used as the eluent at a flow rate of 1.0 mL min$^{-1}$. Data acquisition and calculations were performed using Viscotek OmniSec software version 5.0. Molecular weights were determined based on a conventional calibration curve generated from narrow dispersion polystyrene standards (Agilent and Polymer Laboratories, $M_n = 645–300\,000 \text{g mol}^{-1}$). The samples were filtered over a 0.45 µm PTFE filter prior to injection.

Differential scanning calorimetry (DSC) measurements were conducted to measure the thermal transitions of the obtained furan copolymers. The measurements were performed on a TA-Instru-

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**Figure 5.** (a) Wide-angle X-ray diffraction (WAXD) spectra of the obtained furan-based copolymers from DMFDCA, BHMF and aliphatic diol with a feed ratio of 50:12.5:37.5. (b) POM image of P(FMF-co-DOF) obtained from 50 % DMFDCA, 25 % BHMF, and 25 % 1,8-ODO with a feed ratio of 50:25:25.
ments Q1000 DSC by heating–cooling–heating scans with heating–cooling rates of 10 °C min⁻¹.

Thermogravimetric analysis (TGA) was performed on a TA-Instruments Discovery TGA 5500. The samples were heated at a 10 °C min⁻¹ scan rate in a nitrogen environment. Before the standard TGA measurement, the tested sample was first heated up to 100 °C and then maintained at this temperature for 30 min. to remove the remaining water and solvents in the polymer.

Wide-angle X-ray diffraction (WAXD) patterns of the obtained furan copolymers were recorded on a Bruker D8 Advance diffractometer (Cu Ka radiation, λ = 0.1542 nm) in the angular range of 5–50° (2θ) at room temperature.

Polarized optical microscopy (POM) images were observed by using a Zeiss Axiopt microscope equipped with a Sony DICCC-500 camera for image acquisition. The images were recorded by KS3000 software (Zeiss). The sample preparation was done on a Mettler Toledo FP82HT hot stage with a Mettler FP90 control panel.

**General synthetic procedure for CALB-catalyzed copolymerization with a temperature varied two-stage method**

Based on our previously reported studies,[44] the following temperature varied two-step enzymatic polymerization procedure was applied. As an example, the experimental copolymerization of DMFDCA, BHMF, and diethyl succinate was performed as follows. Predried N435 (20 wt % in relation to the total amount of the monomer) was fed into a 25 mL round bottle under an nitrogen environment. Subsequently, DMFDCA (524 mg, 2.85 mmol), BHMF (730 mg, 5.70 mmol), diethyl succinate (496 mg, 2.85 mmol), and diphenyl ether (6 mL) were added into the flask. In the first step of the reaction, the flask was magnetically stirred in an oil bath and heated to 80 °C for 2 h under a nitrogen atmosphere. Then at the second stage, the pressure was reduced stepwise to 2 mmHg while the reaction temperature was kept at 80 °C for the first 48 h. Finally, the reaction temperature was increased to 95 °C under full vacuum for the last 24 h. After that, the flask was cooled down. Chloroform (20 mL) was added to dissolve the products. N435 was filtered off by normal filtration (Folded filter type 15 Munktell filter) and then washed with chloroform (15 mL) and then dried under vacuum at 200 °C. The obtained polymer was dissolved in chloroform (20 mL) in a 100 mL flask and then poured into a round bottle under nitrogen environment. After centrifugation (2500 rpm, 20 min), the precipitated product was isolated by centrifugation (20 °C for 3 days). Lastly, they were stored under vacuum at room temperature prior to analysis.

The synthesis procedure of the other copolymers was the same as the example above, except using different monomers and feed compositions.

**Furan-based copolymers:**

| Poly(2,5-furandimethylene furanoate-co-butylene furanoate) | [P(FMF-co-BF)]: | H NMR (400 MHz, CDCl₃): δ = 7.10 (2H, m, -CH=, DMFDCA), 6.48 (2H, m, -CH=, BHMF), 5.28 (4H, m, -CO-O-CH₂-C₂H₅, BHMF), 4.38 (4H, m, -CO-O-CH₂-C₂H₅, from 1,4-BDO), 1.90 (4H, m, -CO-O-CH₂-C₂H₅, from 1,4-BDO), 4.61 (s, -CH₂OH, end group from DHMF), 3.92 (s, -O-C₈H₈, end group from DMFDCA), 3.71 ppm (t, -O(CH₂)₄, end group from 1,4-BDO).
| Poly(2,5-furandimethylene furanoate-co-hexamethylene furanoate) | [P(FMF-co-HF)]: | H NMR (400 MHz, CDCl₃): δ = 7.18 (2H, m, -CH=, DMFDCA), 6.48 (2H, m, -CH=, BHMF), 5.28 (4H, m, -CO-O-CH₂-C₂H₅, BHMF), 4.32 (4H, m, -CO-O-CH₂-C₂H₅, from 1,6-HDO), 1.77 (4H, m, -CO-O-CH₂-C₂H₅, from 1,6-HDO), 1.46 (4H, m, -CO-O-CH₂-C₂H₅, from 1,6-HDO), 4.60 (s, -CH₂OH, end group from BHMF), 3.91 (s, -O-C₈H₈, end group from DMFDCA), 3.64 ppm (t, -CH₂OH, end group from 1,6-HDO).
| Poly(2,5-furandimethylene furanoate-co-octamethylene furanoate) | [P(FMF-co-OF)]: | H NMR (400 MHz, CDCl₃): δ = 7.18 (2H, m, -CH=, DMFDCA), 6.48 (2H, m, -CH=, BHMF), 5.28 (4H, m, -CO-O-CH₂-C₂H₅, BHMF), 4.30 (4H, m, -CO-O-CH₂-C₂H₅, from 1,8-DDO), 1.74 (4H, m, -CO-O-CH₂-C₂H₅, from 1,8-DDO), 1.36 (8H, m, -CO-O-CH₂-C₂H₅, from 1,8-DDO), 1.29 (8H, m, -CH₂, from 1,10-DDO), 4.61 (s, -CH₂OH, end group from BHMF), 3.92 (s, -O-C₈H₈, end group from DMFDCA), 3.63 ppm (t, -CH₂OH, end group from 1,8-DDO).
| Poly(2,5-furandimethylene furanoate-co-decamethylene furanoate) | [P(FMF-co-DOF)]: | H NMR (400 MHz, CDCl₃): δ = 7.18 (2H, m, -CH=, DMFDCA), 6.48 (2H, m, -CH=, BHMF), 5.29 (4H, m, -CO-O-CH₂-C₂H₅, BHMF), 4.31 (4H, m, -CO-O-CH₂-C₂H₅, from 1,10-DDO), 1.74 (4H, m, -CO-O-CH₂-C₂H₅, from 1,10-DDO), 1.26 (12H, m, -CH₂, from 1,12-DDO), 4.61 (s, -CH₂OH, end group from BHMF), 3.92 (s, -O-C₈H₈, end group from DMFDCA), 3.64 ppm (t, -CH₂OH, end group from 1,12-DDO).
| Poly(2,5-furandimethylene furanoate-co-2,5-furandimethylene succinate) | [P(FMF-co-FS)]: | H NMR (400 MHz, CDCl₃): δ = 7.13 (2H, m, -CH=, DMFDCA), 6.33 (2H, m, -CH=, BHMF), 5.28 (4H, m, -CO-O-CH₂-C₂H₅, BHMF-DMFDCA), 5.05 (4H, m, -CO-O-CH₂-C₂H₅, BHMF-succinate), 2.63 (4H, m, -O-CO-CH₂, succinate), 4.57 (s, -CH₂OH, end group from BHMF), 3.91 (s, -O-C₈H₈, end group from DMFDCA), 4.13 (m, -OCH₂CH₂, end group from diethyl succinate), 1.23 (t, -OCH₃CH₂, end group from diethyl succinate), 4.45 ppm (s, -CH₂O-CH₂, BHMF ether).
| Poly(2,5-furandimethylene furanoate-co-2,5-furandimethylene adipate) | [P(FMF-co-FA)]: | H NMR (400 MHz, CDCl₃): δ = 7.20 (2H, m, -CH=, DMFDCA), 6.34 (2H, m, -CH=, BHMF), 5.26 (4H, m, -CO-O-CH₂-C₂H₅, BHMF-DMFDCA), 5.01 (4H, m, -CO-O-CH₂-C₂H₅, BHMF-adipate), 2.33 (3H, m, -O-CO-CH₂, adipate), 1.64 (4H, m, -CH₂, adipate), 4.58 (s, -CH₂OH, end group from BHMF), 3.90 (s, -O-C₈H₈, end group from DMFDCA), 4.11 (m, -OCH₂CH₂, end group from diethyl adipate), 1.23 (t, -OCH₃CH₂, end group from diethyl adipate), 4.45 ppm (s, -CH₂O-CH₂, BHMF ether).
Poly(2,5-furandimethylene furanoate-co-2,5-furandimethylene succinate) [P(FMF-co-FMA):] 1 H NMR (400 MHz, CDCl3): δ = 7.21 (2H, m, -CH=, DMFDCA), 6.35 (2H, m, -CH=, BHMF), 5.27 (4H, s, -COO-CH2), BHMF-DMFDCA), 5.02 (4H, s, -COO-CH2, BHMF-sebacate), 2.31 (4H, m, -COO-CH2, sebacate), 1.59 (4H, m, -CH2-, sebacate), 1.26 (8H, m, -CH2-, sebacate), 4.59 (s, -CH2OH end group from BHMF), 3.91 (s, -CH2OH end group from DMFDCA), 4.11 (m, -OCH2CH2 end group from diethyl sebacate), 4.46 ppm (s, -CH2OH, BHMF ether).

Poly(2,5-furandimethylene furanoate-co-2,5-furandimethylene suberate) [P(FMF-co-FMS)]: 1 H NMR (400 MHz, CDCl3): δ = 7.21 (2H, m, -CH=, DMFDCA), 6.35 (2H, m, -CH=, BHMF), 5.27 (4H, s, -COO-CH2), BHMF-DMFDCA), 5.02 (4H, s, -COO-CH2, BHMF-sebacate), 2.31 (4H, m, -COO-CH2, sebacate), 1.59 (4H, m, -CH2-, sebacate), 1.26 (8H, m, -CH2-, sebacate), 4.59 (s, -CH2OH end group from BHMF), 3.90 (s, -OCH2 end group from DMFDCA), 4.11 (m, -OCH2CH2 end group from diethyl sebacate), 4.47 ppm (s, -CH2OH, BHMF ether).

Poly(2,5-furandimethylene furanoate-co-2,5-furandimethylene dodecanedioate) [P(FMF-co-FMDCA):] 1 H NMR (400 MHz, CDCl3): δ = 7.21 (2H, m, -CH=, DMFDCA), 6.35 (2H, m, -CH=, BHMF), 5.27 (4H, s, -COO-CH2), BHMF-DMFDCA), 5.02 (4H, s, -COO-CH2, BHMF-dodecanedioate), 2.31 (4H, m, -COO-CH2, dodecanedioate), 1.60 (4H, m, -CH2CH2-, dodecanedioate), 1.24 (12H, m, -CH2-, dodecanedioate), 4.59 (s, -CH2OH end group from BHMF), 3.91 (s, -OCH2 end group from DMFDCA), 4.10 (m, -OCH2CH2 end group from diethyl dodecanedioate), 4.46 ppm (s, -CH2OH, BHMF ether).

Abbreviations

PEF = poly(ethylene furanoate); PET = poly(ethylene terephthalate); N435 = Novozyme 435; CALB = Candida antarctica lipase B; DMFDCA = dimethyl 2,5-furandicarboxylate; BHMF = 2,5-bis(hydroxymethyl)furan; 1,4-BDO = 1,4-butanediol; 1,6-HDO = 1,6-hexanediol; 1,8-ODD = 1,8-octanediol; 1,10-DDO = 1,10-decanediol; 1,12-DDO = 1,12-dodecanediol; P(FMF-co-BF) = poly(2,5-furandimethylene furanoate-co-butylene furanoate); P(FMF-co-HF) = poly(2,5-furandimethylene furanoate-co-hexamethylene furanoate); P(FMF-co-OF) = poly(2,5-furandimethylene furanoate-co-octamethylene furanoate); P(FMF-co-D) = poly(2,5-furandimethylene furanoate-co-decamethylene furanoate); P(FMF-co-DOP) = poly(2,5-furandimethylene furanoate-co-dodecamethylene furanoate); P(FMF-co-FFM) = poly(2,5-furandimethylene furanoate-co-2,5-furandimethylene succinate); P(FMF-co-FMA) = poly(2,5-furandimethylene furanoate-co-2,5-furandimethylene adipate); P(FMF-co-FMSu) = poly(2,5-furandimethylene furanoate-co-2,5-furandimethylene suberate); P(FMF-co-FMSe) = poly(2,5-furandimethylene furanoate-co-2,5-furandimethylene sebacate); P(FMF-co-FMDCA) = poly(2,5-furandimethylene furanoate-co-2,5-furandimethylene dodecanedioate); ATR-FTIR = attenuated total reflectance–Fourier transform infrared; SEC = size-exclusion chromatography; DSC = differential scanning calorimetry; TGA = thermogravimetric analysis; WAXD = wide-angle X-ray diffraction.

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Conflict of interest

The authors declare no conflict of interest.

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