Chain elongation suppression of cyclic block copolymers in lamellar microphase-separated bulk

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Chain elongation suppression of cyclic block copolymers in microphase-separated bulk was determined quantitatively. Solvent-cast and annealed films are confirmed to show alternating lamellar structure and their microdomain spacing $D$ increases with increasing total molecular weight $M$ according to the relationship $D \propto M^{0.59}$, which agrees quite consistently with the theoretically predicted power law, i.e., $D \propto M^{2/3}$. This result is in contrast to the well-established issue for linear block copolymers, where the relationship $D \propto M^{2/3}$ has been confirmed to hold both experimentally and theoretically. This means that chain elongation of each component block is suppressed considerably, owing to their looped conformation in strongly segregated bulk. © 2004 American Institute of Physics. [DOI: 10.1063/1.1760514]

Block copolymers are fascinating research objectives in both the academic field and the application one, since they exhibit novel properties in condensed systems resulting from forming self-organized periodic structures, called microphase-separated structures.

Cyclic macromolecules have attracted many researchers’ interest since they are conceived to be a kind of “model” polymers with no chain ends, so that it may influence on several properties of polymers considerably. Historically, theoretical works predicted the contracted chain conformation of cyclic molecules compared to linear homologs prior to experimental works.1–3 Later Douglas and Freed predicted possible contraction in dilute solutions by applying renormalization group theory to cyclic molecules,4 while Cates and Deutsch predicted suppression of chain dimension of cyclic molecule in bulk.5 Furthermore, Marko predicted contraction of lamellar microdomain thickness for a ring-shaped block copolymer.6 On the other hand, enormous efforts have been dedicated to preparation and characterization of cyclic homopolymer molecules7–17 and also of cyclic block copolymers.18–21

Throughout these experimental studies, however, the direct evidence of cyclic structure was not shown; moreover, the purity of the cyclic molecules has not been determined quantitatively in most of the works. Among them, Ohtani et al. have proved the cyclic structure directly by using poly(2-vinylpyridine) with detectable junction point by pyrolysis gas chromatography/mass spectroscopy and determined the ring fraction,16 while Lee et al. determined the fractions of cyclic polystyrenes precisely by liquid chromatography at the critical condition method.17 Takano et al. have reported on preparation and characterization of cyclic polystyrenes with short poly (2-tert-butyl-butadiene) sequence and proposed a novel method to prove the purity of cyclic molecule,21 while they also prepared polystyrene-block-polyisoprene cyclic copolymers with three different compositions and determined the purity of cyclic molecules included in the samples accurately.20

Linear block copolymer is known to be elongated considerably along the direction normal to microdomain interface because of strong interaction in between incompatible polymer components.22–27 In contrast to linear homologs, chain elongation of cyclic block copolymers in bulk must be suppressed because they have to adopt looped conformation, their two junction points being tethered on the same microdomain interface. Magnitude of chain compression was calculated by Marko6 following the approach of Leibler,28 assuming Gaussian chain correlation combined with random phase approximation in the strong segregation regime, where actually no chain contraction was predicted comparing domain spacing of $AB$ cyclic copolymer with that of $ABA$ triblock copolymer molecule with the same molecular weight and composition.26 Lescanec et al. experimentally observed domain contraction for two kinds of cyclic block copolymers as a function of $\chi N$, where $\chi$ is the Flory-Huggins interaction parameter and $N$ denotes total degree of polymerization.18 Recently shrinkage of microdomain was also reported for cyclic polystyrene-block-polyisoprene22 and also for cyclic polystyrene-block-polybutadiene29 copolymers; however, suppression of microdomain has not been quantitatively clarified yet. Therefore, molecular weight dependence of the domain spacing of cyclic block copolymers was studied in comparison with the linear homologs to
TABLE I. Molecular characteristics of linear and ring block copolymers.

<table>
<thead>
<tr>
<th></th>
<th>$M_r \times 10^{-3}$</th>
<th>DP</th>
<th>$M_w/M_n$</th>
<th>Purity (%)</th>
<th>$\Phi_\text{c} (%)$</th>
<th>$D$ (nm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>SII-5</td>
<td>31.5</td>
<td>378</td>
<td>1.07</td>
<td>···</td>
<td>50</td>
<td>14.4</td>
</tr>
<tr>
<td>SII-5</td>
<td>34.2</td>
<td>412</td>
<td>1.06</td>
<td>95</td>
<td>49</td>
<td>15.0</td>
</tr>
<tr>
<td>SII-6</td>
<td>69.4</td>
<td>822</td>
<td>1.06</td>
<td>···</td>
<td>53</td>
<td>25.4</td>
</tr>
<tr>
<td>SII-6</td>
<td>67.2</td>
<td>796</td>
<td>1.09</td>
<td>87</td>
<td>53</td>
<td>23.6</td>
</tr>
<tr>
<td>SII-7</td>
<td>101</td>
<td>1216</td>
<td>1.04</td>
<td>···</td>
<td>49</td>
<td>32.6</td>
</tr>
<tr>
<td>SIR-7</td>
<td>100</td>
<td>1184</td>
<td>1.04</td>
<td>79</td>
<td>53</td>
<td>28.9</td>
</tr>
<tr>
<td>SII-8</td>
<td>175</td>
<td>2081</td>
<td>1.06</td>
<td>···</td>
<td>52</td>
<td>46.4</td>
</tr>
<tr>
<td>SIR-8</td>
<td>186</td>
<td>2202</td>
<td>1.05</td>
<td>79</td>
<td>52</td>
<td>41.1</td>
</tr>
</tbody>
</table>

*Total degree of polymerization of samples calculated from molecular weights and volume fractions.

determine the degree of chain suppression exactly in this work using well-characterized samples. Four copolymers with different molecular weight were prepared; their code names are (SI)\text{-}c\text{-}I through (SI)\text{-}c\text{-}IV. Number average molecular weights of (SI)\text{-}c\text{-}I and -II were measured by membrane osmometry while those of (SI)\text{-}c\text{-}III and -IV were determined by combining molecular weights of polystyrene blocks with their mole fractions. The former were estimated by SEC chromatograms of polystyrene after decomposing polyisoprene blocks by osonolysis, while latter were measured by $^1\text{H}$-NMR. Using the same SEC chromatograms, the purity of cyclic molecule was estimated, which comprised coupled polystyrene sequence decomposed from cyclic molecule and small amount of short polystyrene blocks originally on both chain ends of unreacted SIS triblock copolymer whose molecular weight is half of the former. Molecular weights and compositions of linear triblock copolymers, whose code names are SIS-I through SIS-IV, were separately determined. Molecular characteristics of four pair of samples are listed in Table I. From this table, we notice that all the samples have almost the same polystyrene volume fraction $\phi_\text{c}$ of around 0.5 and also that the purity of cyclic molecules is not perfect but sufficiently high to discuss the structural feature of cyclic block copolymer molecule without affecting the results seriously.

Films for morphological observation were cast very slowly from dilute solutions of tetrahydrofuran for four days, followed by heating for a week at 150 °C, which is high enough above glass transition temperature of polystyrene. Microphase-separated structures were observed by transmission electron microscopy and by small angle x-ray scattering; the details of the experiments were reported previously.

As is schematically shown in Fig. 1, if small force $f$ is applied to two opposite points of a block copolymer molecule, the total chain dimension $r$ should be expressed as Eq. (1) (Ref. 30),

$$r \approx R_g^2 f/kT,$$

where $R_g$ denotes radius of gyration of the molecule. The elastic energy $F_e$ per block polymer chain equals $f$ times $r$ so that Eq. (2) holds,

$$F_e = fr \approx kT r^2 / R_g^2.$$

Interfacial energy $F_i$ can be described as

$$F_i = kT \gamma N / r,$$

where $\gamma$ is the surface tension in between two component polymers and is assumed to be large enough to form microphase-separated structure since we are dealing with the system in the strong segregation regime. On the other hand, chain dimension of cyclic molecule can be predicted as a function of $N$ as

$$R_g \approx N^{2/5}.$$

FIG. 1. Schematic comparison of chain elongation of block polymers in lamellar microdomain in between (a) a cyclic block copolymer and (b) a linear triblock copolymer.

![a)](image)

![b)](image)
Introducing the relationship in Eq. (4) into Eq. (2) and replacing domain spacing \( D_c \) for cyclic block copolymer from chain displacement \( r \) because \( D_c \) should be proportional to \( r \) as shown in Fig. 1(b), total free energy \( F \) of forming lamellar structure can be described ignoring numerical prefactor as

\[
F/kT \approx D_c^2/N^{4/5} + N/D_c.
\]

Minimizing \( F \) with respect to \( D_c \), we have the relationship

\[
D_c \approx N^{3/5}.
\]

The relationship in Eq. (6) should be compared with that for linear block copolymer, whose domain spacing \( D_L \) can be expressed by Eq. (7), in the strong segregation regime,

\[
D_L \approx N^{2/3}.
\]

Cyclic block copolymers as well as the corresponding linear ones were confirmed to possess very periodic and qualitatively similar alternating lamellar structure from morphological observation as shown in Fig. 2. Figure 3 compares small-angle x-ray scattering (SAXS) diffraction patterns as an example obtained for one of the cyclic molecules

and the corresponding linear counterpart as a function of the magnitude of the scattering vector \( q \) (= \( 4 \pi \sin \theta /\lambda \)), where \( \lambda \) is the wavelength of x ray and \( 2\theta \) is the scattering angle. It is obvious from this figure that the repeating distance for \( SI \) cyclic molecule estimated by the top curve is shorter than that for a \( SIS \) triblock copolymer obtained from the bottom one. Measured domain spacing \( D_c \) for cyclic copolymers together with those for linear counterparts, \( D_L \), are listed in Table I. From this table one notices that the ratio \( D_c/D_L \) is 0.89 for sample pair IV whose \( \chi N \) is about 210 since total degree of polymerization is about 2100 and \( \chi \) parameter for polystyrene-polysoprene pair is 0.10 at room temperature. This result is consistent with the previously reported one, \(^{18}\) where the \( D_c/D_L \) ratio is 0.91 for polystyrene-block-polydimethylsiloxane cyclic block copolymer with \( \chi N \) of 260.

Using this \( \chi \) value and the total degree of polymerization in Table I, \( \chi N \) for \((SI)_{c}\) was estimated to be ca. 40, this value is large enough compared with theoretically predicted critical value by Marko\(^{6}\) and Borsari\(^{32}\) independently for cyclic block copolymer molecules, ca. 18. Therefore, we can safely assume that all the samples form two-phase structures in the strong segregation regime.

Domain spacing values are plotted against total molecular weights of the samples double logarithmically in Fig. 4 together with those of the linear molecules. It is clear that the difference in \( D \) is getting larger with increasing molecular weight and the exponent \( \alpha \) of the following equation:

\[
D = kM^\alpha
\]

for cyclic molecules is determined to be 0.59, while that for linear molecules is 0.68. The former value is quite consistent with the predicted one as shown in Eq. (6), while the latter agrees well with the well-known two-third power law as is in Eq. (7).\(^ {33}\) This means that the chain dimension along the direction perpendicular to lamellar microdomain interface for a cyclic molecule is meaningfully smaller than that of a linear triblock copolymer molecule with the same molecular weight. This could be attributed to the fact that two blocks of all the cyclic chains must have loop-type conformation in lamellar phase and hence the elongation along this direction can be suppressed as predicted because of the teth-
ering nature. However, it should be noted that the experimentally observed exponent $0.59_4$ is still much larger than the exponent 0.5 in the radius of gyration ($R_g$)–molecular weight ($M$) relationship, $R_g \propto M^{1/2}$, for the unperturbed linear polymer chain.

From this fact, we understand that the polymer chain of cyclic block copolymer in microphase-separated bulk whose both ends are tethered on microdomain interface is still considerably elongated toward the direction normal to the interface because of segregation power in this regime. This relationship, $D = kM^{0.594}$, has been obtained by the counterbalanced effects of (a) block chain elongation due to interfacial segregation power and (b) chain contraction owing to elastic energy from chains whose both ends are tethered on the same microdomain interface.

ACKNOWLEDGMENTS

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31 N. Torikai, Y. Takahashi, A. Takano, and Y. Matsushita (unpublished). Polysoprene block chains for the present block copolymers are preferentially conformed of 1,2- and 3,4-addition microstructures, therefore, interaction parameter against polystyrene is different from that of polystyrene-polysoprene. $\chi$ parameter was determined to be $\chi = 0.035 + 19/7$ by small angle neutron scattering experiments at high temperature for a low-molecular-weight deuterated polystyrene-polysoprene diblock copolymer.